

## Microcontact Printing of Alkanethiols on Copper and Its Application in Microfabrication

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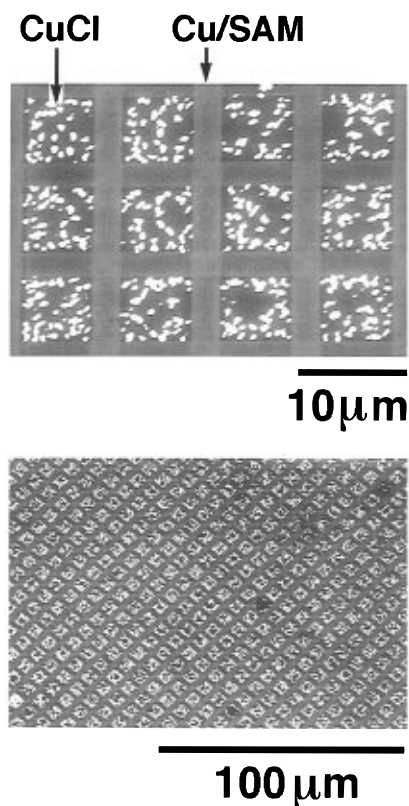
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Microcontact printing ( $\mu\text{CP}$ )<sup>1</sup> is a convenient technique for generating patterned self-assembled monolayers (SAMs)<sup>2</sup> of alkanethiols on gold<sup>1</sup> and silver<sup>3</sup> and of alkylsiloxanes on silicon dioxide and glass.<sup>4,5</sup> Patterned SAMs of hexadecanethiolate on gold and silver can be used directly as ultrathin resists in selective wet etching to fabricate patterned microstructures of these metals with dimensions of  $>0.2\ \mu\text{m}$ .<sup>1,3,6–9</sup> The present study is a natural extension of this technique, since alkanethiols also form organized monolayers on evaporated films of copper.<sup>10</sup> The chemical reactivity of coinage metals increases in the order of  $\text{Au} < \text{Ag} < \text{Cu}$ ; it is possible to find an etchant that dissolves only Cu but not Ag and Au or only Ag but not Au. For example, aqueous  $\text{FeCl}_3$  solution, an etchant that is widely used in semiconductor industry to produce printed circuits of copper,<sup>11</sup> etches silver and gold very slowly or not at all.<sup>3</sup> This ability to carry out “orthogonal etching” offers an opportunity to generate junctions of Cu–Ag, Cu–Au, and Ag–Au by using a multistep procedure that includes metal evaporation, microcontact printing with hexadecanethiol, and selective wet etching.

- (1) Kumar, A.; Whitesides, G. M. *Appl. Phys. Lett.* **1993**, *63*, 2002.  
 (2) Whitesides, G. M.; Laibinis, P. E. *Langmuir* **1990**, *6*, 87. Dubois, L. H.; Nuzzo, R. G. *Annu. Rev. Phys. Chem.* **1992**, *43*, 437.  
 (3) Xia, Y.; Kim, E.; Whitesides, G. M. *J. Electrochem. Soc.*, in press.  
 (4) Xia, Y.; Mrksich, M.; Kim, E.; Whitesides, G. M. *J. Am. Chem. Soc.* **1995**, *117*, 9576.  
 (5) Jeon, N. L.; Nuzzo, R. G.; Xia, Y.; Mrksich, M.; Whitesides, G. M. *Langmuir* **1995**, *11*, 3024.  
 (6) Kumar, A.; Biebuyck, H.; Whitesides, G. M. *Langmuir* **1994**, *10*, 1498.  
 (7) Xia, Y.; Whitesides, G. M. *J. Am. Chem. Soc.* **1995**, *117*, 3274.  
 (8) Xia, Y.; Whitesides, G. M. *Adv. Mater.* **1995**, *7*, 471.  
 (9) Wilbur, J. L.; Kim, E.; Xia, Y.; Whitesides, G. M. *Adv. Mater.* **1995**, *7*, 649.  
 (10) Laibinis, P. E.; Whitesides, G. M. *J. Am. Chem. Soc.* **1992**, *114*, 9022.  
 (11) Glang, R.; Gregor, L. V. In *Handbook of Thin Film Technology*; Maissel, L. I., Glang, R., Eds.; McGraw-Hill: New York, 1970; Chapter 7.  
 (12) Copper (99.999%, Aldrich) films were prepared by e-beam evaporation onto Ti-primed (99.99%, Aldrich; 2.5 nm thick) Si(100) wafers (N/phosphorous doped, Silicon Sense, Nashua, NH). Microcontact printing was carried out according to published procedures.<sup>6</sup> Etchings of copper were conducted at room temperature; etching solutions were stirred at 300 rpm. We measured the thickness of hexadecanethiolate SAMs on copper using ellipsometry. SAMs formed by dipping copper films into an  $\sim 2\ \text{mM}$  hexadecanethiol solution in ethanol for  $\sim 5\ \text{s}$  had a thickness of  $\sim 58\ \text{\AA}$  (obviously, they were bilayers, for the reasons that are, however, not clear at the present time<sup>13</sup>); SAMs formed by  $\mu\text{CP}$  using a flat PDMS stamp had a thickness of  $\sim 20\text{--}26\ \text{\AA}$  (that is, they were very close to monolayers).  
 (13) Keller, H.; Simak, P.; Schrepp, W.; Dembowski, J. *Thin. Solid Films* **1994**, *244*, 799. Schlenoff, J. B.; Li, M.; Ly, Hiep. *J. Am. Chem. Soc.* **1995**, *117*, 12528.

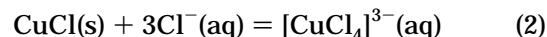


**Figure 1.** SEMs of copper films ( $\sim 50\ \text{nm}$  thick) that were patterned with SAMs of hexadecanethiolate and etched in an aqueous  $\text{FeCl}_3$  solution for  $\sim 2\text{--}3\ \text{s}$ . The microparticles (bright dots on SEMs) were  $\text{CuCl}$  that formed in situ on the bare regions of copper during etching.

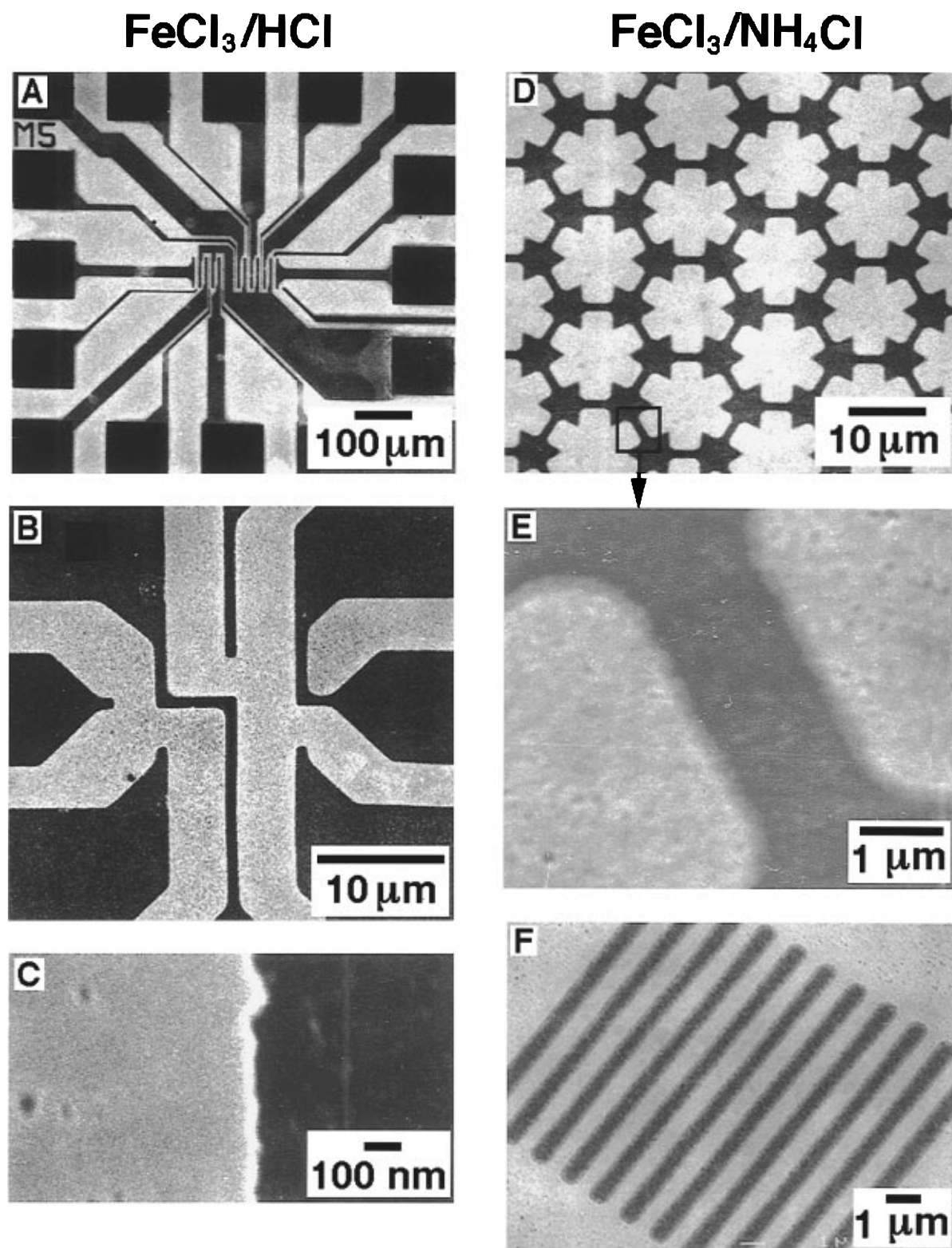
Figure 1 shows scanning electron micrographs (SEMs) of copper samples ( $50\ \text{nm}$  thick) that had been patterned with SAMs of hexadecanethiolate and etched in an aqueous  $\text{FeCl}_3$  solution ( $0.012\ \text{M}$ ) for  $2\text{--}3\ \text{s}$ .<sup>12</sup> The microparticles ( $\sim 0.5\ \mu\text{m}$  in size) on the bare regions of copper are  $\text{CuCl}$  (shown by energy-dispersed spectroscopy, EDS). They were formed in situ during the dissolution of bare copper:



Addition of  $\text{HCl}$  or  $\text{NH}_4\text{Cl}$  to the etching solution decreased the rate of etching and helped to dissolve the  $\text{CuCl}$  precipitate.<sup>14</sup>



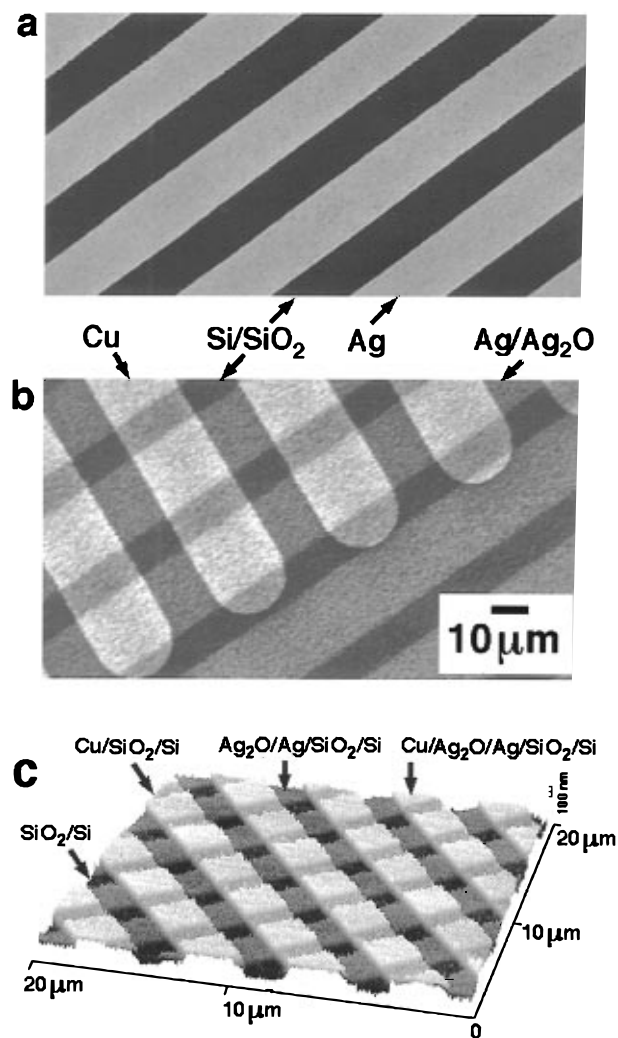
- (14) Cotton, F. A.; Wilkinson, G. *Advanced Inorganic Chemistry*, 5th ed.; John Wiley & Sons: New York, 1988; p 759.  
 (15) Biebuyck, H.; Whitesides, G. M. *Langmuir* **1994**, *10*, 4581.  
 (16) Dirske, T. P. In *Zinc-Silver Oxide Batteries*; Fleischer, A., Lander, J. J., Eds.; John Wiley & Sons: New York, 1971; p 100.  
 (17) UV Clean, Model 135500, Boekel Instruments, PA.  
 (18) The XPS of freshly prepared silver films only showed the characteristic peak of  $\text{Ag}(0)$  ( $3d$ ,  $368.08\ \text{eV}$ ). When the surfaces of these silver films were oxidized under UV/ozone, the peak of  $3d$  shifted to lower energies:  $368.03\ \text{eV}$  ( $1\ \text{min}$ );  $368.02\ \text{eV}$  ( $5\ \text{min}$ ), and  $367.65\ \text{eV}$  ( $10\ \text{min}$ ). The characteristic peak ( $3d$ ) of  $\text{Ag}_2\text{O}$  is at  $367.65\ \text{eV}$ . In fabricating arrays of junctions, the surfaces of silver films were oxidized for  $\sim 6\ \text{min}$ .  
 (19) When we were preparing this revised paper, we noticed that a similar work ( $\mu\text{CP}$  of hexadecanethiol on copper and selective etching) was just published: Moffat, T. P.; Yang, H. *J. Electrochem. Soc.* **1995**, *142*, L220. The etchant used by these authors was an aqueous solution containing  $\text{H}_2\text{O}_2$  and  $\text{HCl}$ .



**Figure 2.** SEMs of copper films ( $\sim 50$  nm thick) that were patterned with SAMs of hexadecanethiolate and etched in aqueous solutions containing  $\text{FeCl}_3/\text{HCl}$  (A–C) or  $\text{FeCl}_3/\text{NH}_4\text{Cl}$  (D–F) for  $\sim 20$  s. Addition of HCl or  $\text{NH}_4\text{Cl}$  to the  $\text{FeCl}_3$  solution helped to dissolve the precipitates of  $\text{CuCl}$  and to generate clean structures of copper. The bright regions are copper covered by the SAM, and the dark regions are  $\text{Si}/\text{SiO}_2$  where the underivatized copper has dissolved.

Figure 2A–C shows SEMs of test patterns of copper (at three different magnifications) that were fabricated by  $\mu\text{CP}$  with hexadecanethiol, followed by etching of the patterned samples in an aqueous solution containing  $\text{FeCl}_3$  (0.012 M) and HCl (0.4–0.8 M) for  $\sim 20$  s. The

edge resolution of copper structures is  $\sim 100$  nm (Figure 2C) when using a poly(dimethylsiloxane) (PDMS) stamp cast from lines made in a film of photoresist (Microposit 1813, Shipley) using photolithography; the edge resolutions are  $\sim 50$  and  $\sim 20$  nm for the systems of  $\text{Au}/\text{C}_{16}$ -



**Figure 3.** (a) SEM image of silver lines ( $\sim 50$  nm thick) that were generated by a combination of  $\mu$ CP and selective etching in  $S_2O_3^{2-}$ /ferri/ferrocyanide solution for  $\sim 15$  s. (b) SEM image of Ag/Ag<sub>2</sub>O/Cu junction structures. Samples of (a) were oxidized under ozone for  $\sim 6$  min, films of copper ( $\sim 50$  nm) were evaporated and patterned by  $\mu$ CP, followed by selective etching in aqueous FeCl<sub>3</sub>/HCl solutions. (c) AFM image of (b) to show 3-D profile.

SH<sup>15</sup> and Ag/C<sub>16</sub>SH,<sup>3</sup> respectively. Figure 2D–F shows SEMs of other test patterns of copper that were generated using the combination of  $\mu$ CP with hexadecanethiol and selective wet etching in another aqueous solution containing FeCl<sub>3</sub> (0.012 M) and NH<sub>4</sub>Cl (0.8 M) for  $\sim 20$  s. The smallest features of copper that have been fabricated using the present procedure are parallel lines of copper that are  $\sim 0.6$   $\mu$ m in width and are separated by  $\sim 0.6$   $\mu$ m (Figure 2F).

Since aqueous FeCl<sub>3</sub>/HCl solutions only dissolve bare copper, not bare silver, we have been able to fabricate arrays of Ag/Ag<sub>2</sub>O/Cu junctions by a four-stage procedure. (i) Generate silver lines (50 nm thick) on Si/SiO<sub>2</sub> using a combination of  $\mu$ CP with hexadecanethiol and selective etching of silver in aqueous solutions containing S<sub>2</sub>O<sub>3</sub><sup>2-</sup> and ferri/ferrocyanide (Figure 3a).<sup>3</sup> (ii) Form insulating layer of Ag<sub>2</sub>O<sup>16</sup> on the surface of these silver lines using UV/ozone treatment.<sup>17,18</sup> (iii) Evaporate a copper film (50 nm thick) over the whole surface. (iv) Pattern the copper surface with lines of hexadecanethiolate perpendicular to the silver lines and etch in aqueous FeCl<sub>3</sub>/HCl solutions (Figure 3b). Each cross point of silver and copper lines represents a junction of metal/insulator/metal. Figure 3c is an AFM image of b to show the profiles of the junctions.

We have demonstrated that  $\mu$ CP could be used to generate patterned SAMs of alkanethiolates on the surface of evaporated thin films of copper.<sup>19</sup> These patterned SAMs were effective resists in protecting the underlying copper from dissolving in FeCl<sub>3</sub>/HCl and FeCl<sub>3</sub>/NH<sub>4</sub>F solutions. The capability to etch Cu independently of Ag and Au adds another level of useful control to the ability to fabricate microstructures by using SAMs and  $\mu$ CP.

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